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# Grinding of Aluminium-Based Metal Matrix Composites Reinforced with Al<sub>2</sub>O<sub>3</sub> or SiC Particles

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This paper presents results obtained from the grinding of aluminium-based metal matrix composites reinforced with either aluminium oxide  $(Al_2O_3)$  or silicon carbide (SiC) particles using grinding wheels made of SiC in a vitrified matrix or diamond in a resin-bonded matrix. The study used grinding speeds of 1100–2200 m min<sup>-1</sup>, a grinding depth of 15  $\mu$ m for rough grinding and 0.1–1  $\mu m$  for fine grinding, a crossfeed of 3 mm and 1 mm for rough and fine grinding, respectively, while maintaining a constant table feedrate of 20.8 m min<sup>-1</sup>. Surface integrity of the ground surfaces and subsurfaces was analysed using a scanning electron microscope and a profilometer. Grinding using a 3000-grit diamond wheel at depths of cut of 1 µm and 0.5 µm produced ductile streaks on the  $Al_2O_3$  particles and the SiC particles, respectively. There was almost no subsurface damage except for rare cracked particles when fine grinding with the diamond wheel.

**Keywords:**  $Al_2O_3$  particle; Ductile streaks; Grinding; Metal matrix composites; SEM analysis; SiC particles; Surface integrity

## 1. Introduction

Aluminium-based metal matrix composites (MMCs) reinforced with ceramic particles are advanced materials known for their good damping properties, high specific strength, and high wear resistance. Methods to produce these composites and studies on their mechanical properties have gained popularity [1]. MMCs are increasingly used in astronautic, automobile, and military industries. In addition, the sporting goods industry has also been in the forefront of MMCs development capitalising on the materials' high specific properties. There is also a growing interest in the shipping industry [2].

Reports on machining of aluminium-based MMCs reinforced with ceramic particles [3–6] are still scarce. Despite many advantages, full implementation of MMCs is cost-prohibitive. This is partially due to the poor machinability of the materials. Although near-net-shape MMC components can be produced, final finishing is still required for obtaining designed final dimensions and required surface finish. Significant cost and fabrication problems, including machining, must be overcome for the successful application of these composites. Surface finish and surface integrity are important for surface sensitive parts subjected to fatigue. Subsurface damage due to the machining of MMCs results from conventional and unconventional processes, therefore, finishing processes such as grinding are used to improve the surface integrity of machined MMCs [7,8].

Work on grinding silicon and germanium revealed that ductile chips could be obtained by properly controlling the depth of cut [9]. A model for the critical depth associated with ductile-mode machining has been proposed [10]. It was reported that by having a critical depth of cut and with flattened grains slightly protruding from the surface of the grinding wheel, flawless machining, free of brittle fracture, was possible [11]. Evidence of plastic flow with aluminium oxidide (Al<sub>2</sub>O<sub>3</sub>), silicon carbide (SiC), and silicon nitride (Si<sub>3</sub>N<sub>4</sub>) was shown and a model based on the combination of two theories was proposed [12]. A hundred per cent ductile-mode mirror grinding of SiC could be achieved by using a non-ultraprecision machine with a 1  $\mu$ m in-feed (depth of grinding) which would produce large mirror surfaces comparable to well-polished mirrors [13,14].

However, reports on the ductile-mode machining of aluminium-based MMCs reinforced with ceramic particles are still very scarce. Therefore, further studies on the ductile-mode machining of these materials to obtain damage-free surfaces are required for the application of these materials. This paper presents results obtained from the grinding of aluminium-based MMCs reinforced with either SiC or  $Al_2O_3$  particles. The issues discussed are surface roughness, grinding force, type and size of the abrasives, grinding conditions, ductile streaks on  $Al_2O_3$  and SiC particles and the consequential subsurface integrity.

## 2. Experiments

The details of the ground MMC specimens  $2618/Al_2O_3/10p$  (10 vol%  $Al_2O_3$ ),  $2618/Al_2O_3/20p$  (20 vol%  $Al_2O_3$ ), and

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Table 1. The metal matrix composites machined.

| Material      | 2618/Al <sub>2</sub> O <sub>3</sub> /10p  | 2618/Al <sub>2</sub> O <sub>3</sub> /20p  | A359/SiC/10p-T6  |
|---------------|---|---|--|
| Matrix        | 2618 aluminium alloy  | 2618 aluminium alloy  | A359 aluminium alloy: 8.96 wt% Si,<br>0.18 Fe, 0.53 Mg, balance Al   |
| Reinforcement | 10 vol% $Al_2O_3$ particulate particle size: 9.3 $\mu$ m                                  | 20 vol% $Al_2O_3$ particulate particle size: 21 $\mu$ m                                   | 10 vol% SiC particulate<br>mean 13 $\mu$ m, aspect ratio 1.5:1<br>97% particle <25 $\mu$ m<br>6% <5 $\mu$ m  |
| Process       | Extrusion at 420–430°C<br>Extrusion ratio: 20:1<br>Extrusion speed: 3 m min <sup>-1</sup> | Extrusion at 420–430°C<br>Extrusion ratio: 20:1<br>Extrusion speed: 3 m min <sup>-1</sup> | Pouring temperature: 700–710°C<br>Hot isostatic pressing by heating at<br>550°C, isostatically pressed at 150<br>MPa, oven-cooled to 300°C, then air-<br>cooled<br>Solution heat-treated at 540°C, water<br>quenched, peak aged at 155°C |

A359/SiC/10p-T6 (10 vol% SiC) are shown in Table 1. The MMCs are cast aluminium alloys reinforced with  $Al_2O_3$  or SiC particles. A359/SiC/10p-T6 was hot-isostatically pressed and aged to enhance the matrix properties.

Table 2 shows the details of the grinding conditions and the grinding wheels used. A resin-bonded diamond grinding wheel was used to fine grind the MMC specimens at low, medium, and high grinding speeds, respectively. The grit size was 3000 (5  $\mu$ m average grain size). Some 2618/Al<sub>2</sub>O<sub>3</sub>/10p and 2618/Al<sub>2</sub>O<sub>3</sub>/20p workpieces were also rough ground with an 80-grit vitrified-bond SiC grinding wheel at low, medium, and high grinding speeds, respectively.

Grinding experiments were carried out on an Okamoto precision surface-grinding machine. An inverter was attached to the machine spindle motor so that the main spindle of the machine was capable of changing speed. A dynamometer was mounted on the table of the grinding machine to measure the grinding forces. The dynamometer was connected to charge amplifiers, and the measured grinding forces were recorded using a chart recorder. The grinding force reported here is the force perpendicular to a ground surface. A SiC wheel mounted on a brake-controlled trueing device and a single diamond dresser were used for trueing the grinding wheels. SiC and WA dressing sticks were used for dressing the SiC and diamond grinding wheels, respectively. Dressing was carried out before every grinding experiment.

The surface roughness of the ground MMCs was measured in the direction perpendicular to the grinding direction using a profilometer. The cut-off was 0.8 mm and evaluation length was 4 mm. The average value was calculated from three values measured on each ground surface.

Surface integrity of the machined surfaces and subsurface damage were assessed using a scanning electron microscope (SEM). The samples were observed in the as-machined condition. Some samples were observed again after being etched in Keller's etchant (190 ml water, 5 ml nitric acid, 3 ml hydrochloric acid, 2 ml fluoric acid) to dissolve the smeared aluminium on the surfaces. Selected samples were sectioned, moulded, hand ground, polished, and then etched to show the microstructure at the subsurface.

| Table 2. Grinding | conditions | and | grinding | wheels | used. |
|-------------------|------------|-----|----------|--------|-------|
|-------------------|------------|-----|----------|--------|-------|

| Attributes     |   | Rough grinding of 2618/Al <sub>2</sub> O <sub>3</sub> /10p | Fine grinding of 2618/Al <sub>2</sub> O <sub>3</sub> /20p | Finding grinding of A359/SiC/10p-T6        |
|----------------|---|--|---|--|
| Grinding wheel | Grain   | Green SiC  | Diamond   | Diamond                                    |
|                | Grit size   | 80   | 3000  | 3000                                       |
|                | Bond  | Vitrified bond   | Resin bond  | Resin bond                                 |
|                | Diameter (mm)   | 350  | 350   | 350  |
|                | Width (mm)  | 38   | 10  | 10   |
| Dressing stick | Grain   | Green SiC  | WA  | WA   |
|                | Grit size   | 60   | 320   | 320  |
| Grinding speed | Low (m min <sup>-1</sup> )<br>Medium (m min <sup>-1</sup> )<br>High (m min <sup>-1</sup> )<br>Depth of cut ( $\mu$ m)<br>Feedrate (m min <sup>-1</sup> )<br>Cross-feed (mm) | 1100<br>1650<br>2200<br>15<br>20.8<br>3                    | 1100<br>1650<br>2200<br>1<br>20.8<br>1                    | 1100<br>-<br>0.1, 0.2, 0.5, 1<br>20.8<br>1 |



Fig. 1. Surface roughness of ground  $2618/Al_2O_3/10p$  and  $2618/Al_2O_3/20p$ . Rough grinding (80-grit SiC wheel; depth of cut: 15  $\mu$ m; cross-feed: 3 mm). Fine grinding (3000-grit diamond wheel; depth of cut: 1  $\mu$ m; cross-feed: 1 mm).

## 3. Results and Discussion

The surface roughness values of rough- and fine-ground  $2618/Al_2O_3/10p$  (10 vol%  $Al_2O_3$ ) and  $2618/Al_2O_3/20p$  (20 vol%  $Al_2O_3$ ) are shown in Fig. 1. Values of the maximum grinding force measured during the rough and fine grinding experiments are shown in Fig. 2. Figures 3 and 4 show the top and cross-section of rough-ground surfaces, whereas Figs 5 to 8 show those of fine-ground surfaces.

As shown in Fig. 1, the surface finish Ra of the fine-ground 2618/Al<sub>2</sub>O<sub>3</sub>/10p (10 vol% Al<sub>2</sub>O<sub>3</sub>) was better than that of fineground 2618/Al<sub>2</sub>O<sub>3</sub>/20p (20 vol% Al<sub>2</sub>O<sub>3</sub>). However, the Ra of rough-ground 2618/Al<sub>2</sub>O<sub>3</sub>/20p. This demonstrates the effects of the Al<sub>2</sub>O<sub>3</sub> particles of the ground samples on the performance of the SiC and diamond grinding wheels. For the rough-ground samples, the Ra values were scattered in the range 0.15–0.70  $\mu$ m. A narrower range of 0.20–0.35  $\mu$ m was achieved for the fine-ground samples. Surfaces ground by the 80-grit SiC wheel at speeds of 1100 and 1650 m min<sup>-1</sup> at depth of cut of 15  $\mu$ m had roughness values close to those of surfaces produced by the 3000-grit diamond wheel at depth of cut of 1  $\mu$ m. This



Fig. 2. Forces for grinding  $2618/Al_2O_3/10p$  and  $2618/Al_2O_3/20p.$  Conditions as Fig. 1.



Fig. 3. SEM micrograph of ground  $2618/Al_2O_3/20p$  surface. Rough grinding (80-grit vitrified-bond SiC wheel; grinding speed: 2200 m min<sup>-1</sup>; depth of cut: 15  $\mu$ m; cross-feed: 3 mm; feedrate: 20.8 m min<sup>-1</sup>).



Fig. 4. Subsurface of ground  $2618/Al_2O_3/20p$ . The arrows show cracks of commonly found  $Al_2O_3$  particles. Rough grinding, conditions as Fig. 3.

was due to the smearing of the aluminium matrix. Smearing of aluminium on the ground surfaces was seen for the rough grinding, but was negligible for the fine grinding because all the  $Al_2O_3$  particles of the ground surfaces were clearly visible when observed with the SEM.

The maximum grinding force decreased with increasing grinding speed for rough grinding, but increased with increasing grinding speed for fine grinding. This could be due to several reasons such as the different abrasives, grit sizes and depths of cut used for the rough- and fine-grinding experiments, and the thermal-induced softened matrix at high speeds for rough grinding, etc. For example, because the depth of cut was 15  $\mu$ m for rough grinding of the Al<sub>2</sub>O<sub>3</sub> particles (particle size: 9.3 or 21  $\mu$ m), more heat was generated in the deformation zone. This softened the matrix at the higher grinding speed and lowered the grinding force component perpendicular to the ground surface. In the case of fine grinding, because the depth

5 μm

**Fig. 5.** SEM micrograph of ground  $2618/Al_2O_3/20p$  surface. Fine grinding (3000-grit resin-bond diamond wheel; grinding speed: 1100 m min<sup>-1</sup>; depth of cut: 1  $\mu$ m; cross-feed: 1 mm; feedrate: 20.8 m min<sup>-1</sup>).



Fig. 6. Subsurface of ground  $2618/Al_2O_3/20p$ . Find grinding, conditions as Fig. 5.



Fig. 7. Subsurface of ground  $2618/Al_2O_3/20p$ . The arrow shows the rare crack of an  $Al_2O_3$  particle. Fine grinding, conditions as Fig. 5, but grinding speed: 2200 m min<sup>-1</sup>.



Fig. 8. SEM micrograph of ground A359/SiC/10p-T6 surface. Fine grinding, conditions as Fig. 5, but depth of cut: 0.5  $\mu$ m.

of cut was 1  $\mu$ m, the thermal effect was presumably negligible. Further investigation is required to understand better the micromachining mechanism of both the soft matrix and the hard, brittle particles at the same time. However, from Fig. 2, it can be seen clearly that the grinding force required for grinding the MMCs with 20 vol% Al<sub>2</sub>O<sub>3</sub> particles is usually higher than that for grinding the MMCs with 10 vol% Al<sub>2</sub>O<sub>3</sub> particles.

Figures 3 and 4 show that cracks in the  $Al_2O_3$  particles occur at the surface and under the rough ground surfaces. As mentioned above, surfaces rough ground at speeds of 1100 and 1650 m min<sup>-1</sup> have roughness values close to those of fine-ground surfaces. However, SEM pictures show that the surface topographies of the rough- and fine-ground surfaces having very similar surface roughness values, are significantly different. No  $Al_2O_3$  particles were seen on the rough-ground surfaces, except some small holes showing fractured  $Al_2O_3$  particles, as shown in Fig. 3. Almost the whole surfaces were smeared with the soft aluminium matrix. Some of aluminium chips were back-transferred onto the top of the surfaces.

SiC wheels are much cheaper than diamond wheels. The cost ratio is roughly 1:10–20. Because the depth of cut and cross-feed used were 15 and 3 times those for fine grinding, using a depth of cut 1  $\mu$ m, respectively, the specific material removal rate was 45 times that of the fine grinding using the diamond wheel. The grinding time was also much shorter than for fine grinding. Hence, the potential of using SiC wheels, at least for rough grinding, is high. Rough grinding parameters and dressing frequency should be optimised to make rough grinding using SiC wheels more attractive.

As shown in Fig. 5, grinding of  $2618/Al_2O_3/20p$  (20 vol%  $Al_2O_3$ ) using the fine-grit diamond wheel at 1  $\mu$ m in-feed (depth of grinding) produced visible ductile streaks on the  $Al_2O_3$  particles. Both the matrix and the  $Al_2O_3$  particles were removed by micromachining because the ductile grinding marks were clearly seen on the  $Al_2O_3$  particles. There were no cracks or defects found on the ground surfaces. There was almost no subsurface damage as shown in Fig. 6, except a very rare cracked particle as shown in Fig. 7.

The diamond wheel also produced A359/SiC/10p-T6 (10 vol% SiC) surfaces with few SiC-particle-related defects. A very thin smearing layer of aluminium and partially hidden SiC particles was observed on the ground surfaces. When lightly etched with Keller's etchant, the ductile-mode ground

surfaces of the SiC particles can be seen. As shown in Fig. 8, with a depth of cut of 0.5  $\mu$ m, ductile-mode grinding of SiC particles was observed.

#### 4. Summary

Diamond grinding experiments were performed on aluminiumbased MMCs reinforced with SiC or  $Al_2O_3$  particles. The potential of using SiC wheels at least for rough grinding of alumina/aluminium composites is high, because SiC grains are harder than the  $Al_2O_3$  reinforcing particles and are much less expensive than diamond grains. Rough grinding with a SiC wheel followed by fine grinding with a fine-grit diamond wheel is recommended for the grinding of alumina/aluminium composites. Grinding using a 3000-grit diamond wheel at depths of cut of 1  $\mu$ m and 0.5  $\mu$ m produced many ductile streaks on the  $Al_2O_3$  particles and the SiC particles, respectively.

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