

Direct Laser-Patterned Micro-Supercapacitors from Paintable MoS₂ Films

Liujun Cao, Shubin Yang,* Wei Gao, Zheng Liu, Yongji Gong, Lulu Ma, Gang Shi, Sidong Lei, Yunhuai Zhang, Shengtao Zhang,* Robert Vajtai, and Pulickel M. Ajayan*

Micrometer-sized electrochemical capacitors have recently attracted attention due to their possible applications in micro-electronic devices. Here, a new approach to large-scale fabrication of high-capacitance, two-dimensional MoS_2 film-based microsupercapacitors is demonstrated via simple and low-cost spray painting of MoS_2 nanosheets on Si/SiO₂ chip and subsequent laser patterning. The obtained microsupercapacitors are well defined by ten interdigitated electrodes (five electrodes per polarity) with 4.5 mm length, 820 µm wide for each electrode, 200 µm spacing between two electrodes and the thickness of electrode is ~0.45 µm. The optimum MoS_2 -based micro-supercapacitor exhibits excellent electrochemical performance for energy storage with aqueous electrolytes, with a high area capacitance of 8 mF cm⁻² (volumetric capacitance of 178 F cm⁻³) and excellent cyclic performance, superior to reported graphene-based micro-supercapacitors. This strategy could provide a good opportunity to develop various micro-/nanosized energy storage devices to satisfy the requirements of portable, flexible, and transparent micro-electronic devices.

L. J. Cao, Dr. S. B. Yang, Dr. Z. Liu, L. L. Ma, G. Shi, S. D. Lei, Dr. R. Vajtai, Prof. P. M. Ajavan Department of Mechanical **Engineering & Materials Science Rice University** Houston, Texas 77005, USA E-mail: sy13@rice.edu; ajayan@rice.edu L. J. Cao, Prof. Y. H. Zhang, Prof. S. T. Zhang School of Chemistry and Chemical Engineering **Chongging University** Chongqing 400044, China E-mail: stzhang@cqu.edu.cn Y. J. Gong, Prof. P. M. Ajayan Department of Chemistry **Rice University** Houston, Texas 77005, USA Dr. W. Gao Los Alamos National Laboratory Los Alamos, NM, 87545, USA

DOI: 10.1002/smll.201203164



1. Introduction

Micro-electrochemical capacitors are important complements to batteries due to their high power density.^[1–3] To date, two kinds of capacitive behaviors have been observed in materials used as electrodes in micro-electrochemical capacitors based on their different energy storage mechanisms. One is the electrochemical double layer capacitance (EDLC) related to the double layer formation at the interfaces between electrolyte and electrode surface,^[7] while the other is pseudo-capacitance related to the fast and reversible redox reactions occurring in the electrodes.^[4,5] Very recently, various carbons including carbide-derived carbon,^[6] activated carbon,^[7] onion-like carbon,^[8] carbon nanotubes^[9-11] and graphene sheets (or reduced graphene oxide sheets)^[12,13] have been widely used as electrode materials for EDLC based micro-sized or thin-film supercapacitors. In this regard, graphene is one of the most attractive materials due to its high surface area^[15] and exceptional mechanical properties and high electrical conductivity.^[14] Unfortunately,

full papers

the low area capacitance of graphene micro-devices^[12] $(0.5 - 3 \text{ mF cm}^{-2})$ based on EDLC partially hampers its broad use in high power applications.

MoS₂ nanosheets, a graphene-like 2D material, have shown unique structural and electronic properties.^[16-19,29-35] In a single MoS₂ nanosheet, the Mo atomic layer is sandwiched between two sulfur layers via covalent bonding. And their multi-layers are stacked by van der Waals force, allowing easy intercalation of foreign ions (H⁺, Li⁺). More importantly, the center Mo atoms possess a range of oxidation states from +2 to +6, promising a typical pseudo-capacitance behavior with very high specific capacitance for MoS₂ nanosheets, similar to that of RuO_2 the theoretical capacitance ~1000 F g⁻¹. Moreover, several procedures such as hydrothermal processes,^[20] intercalation-exfoliation method^[21,22] and liquid phase exfoliation have been developed to prepare MoS₂ nanosheets in bulk quantities.^[23,24] Combined with good fabrication technologies from nanosheets to films, MoS₂ nanosheets should be a promising pseudo-capacitance material for high performance micro-supercapacitors.

Here, we demonstrate a new approach for large-scale fabrication of 2D MoS_2 film based micro-supercapacitors *via* simple spray painting of MoS_2 nanosheets to form thin films and subsequent laser patterning. The resulting MoS_2 micro-supercapacitors are consisting of well-defined finger-like electrodes built from large number of curly MoS_2 nanosheets; the structure of the film facilitates the access of the ions to the electrode surface. The optimized micro-supercapacitor

exhibits excellent electrochemical performances and energy storage using aqueous electrolytes, it has high area capacitance of 8 mF cm⁻² and excellent cyclic performance, superior to reported graphene-based micro-supercapacitors.

2. Results and Discussion

As illustrated in Figure 1, 2D MoS₂ film based micro-supercapacitors were fabricated via laser patterning of painted MoS₂ films. First, two types of free-standing MoS₂ nanosheets were synthesized via an anionic sulfate surfactant assisted hydrothermal process and a liquid exfoliation approach; the as-prepared MoS₂ nanosheets were denoted as h-MoS₂ and e-MoS₂, respectively. Both of the MoS₂ nanosheet samples were then dispersed in mixed solutions of ethanol and water (ethanol/water volume rate = 4:1) to form inks for painting (for details, see Supporting Information). Subsequently, the stable suspensions were spray painted onto Si\SiO₂ wafers and MoS₂ films with uniform thickness were generated after the solvent evaporated (Figure 1A, steps 1 and 2). A fingerelectrode pattern was defined by laser patterning (Figure 1A, step 3), and the obtained pattern electrode was assembled into an electrochemical test cell (Figure 1C). The photograph of a typical painted and patterned film (Figure 1B) shows a well-defined arrangement and the separation between the electrodes; the micro-device was constructed with 10 interdigitated electrodes (4.5 mm length, 820 µm wide for each electrode) with appropriate thickness (~0.45 µm, five electrodes



Figure 1. (A) Fabrication of MoS_2 film micro-supercapacitors in three steps: 1) deposition of gold electrical conductive pads on the standard silicon chips for use as current collectors, 2) masking and spray painting MoS_2 ink on the chips, 3) patterning interdigitated finger electrodes (five electrodes per polarity) on the active material area by laser cutting process. (B) A typical photograph of the as-prepared MoS_2 film micro-supercapacitor. Each of the finger electrodes is 4.5 mm long, 820 μ m wide and the spacing between two finger electrodes is 200 μ m. (C) Top view of the electrochemical testing device. Cu wire was used for electrical contacts. The black area is the MoS_2 active material including patterned interdigitated finger electrodes and pads, and the white area between them is the separator which is filled with electrolyte during the electrochemical tests.



Figure 2. Characterization of representative $h-MoS_2$ finger electrodes and $h-MoS_2$ nanosheets. (A) Typical SEM image of a finger-like $h-MoS_2$ electrode (solid line area), the space was generated by laser cutting (dot line area). (B) SEM image of $h-MoS_2$ nanosheets that constructed the $h-MoS_2$ film micro-supercapacitor. (C) HRTEM image of $h-MoS_2$ nanosheets demonstrated to be 2~7 atomic layers. (D) X-ray diffraction (XRD) patterns of $h-MoS_2$ nanosheets.

per polarity (Figure S1 in Supporting Information). This structure of the device is demonstrated by the typical SEM image of one finger electrode in **Figure 2**(A), where the MoS_2 nanosheets are well controlled as a result of the patterning process (Figure S2). Note that such a simple strategy allows us to prepare various micro-electrodes without use of any organic binders. (For details, see Supporting Information).

The morphology and the crystalline structure of h-MoS₂ and e-MoS₂ nanosheets sprayed to form the MoS₂ film electrodes were investigated via scanning electron microscopy (SEM) and transmission electron microscopy (TEM). In case of h-MoS₂ nanosheets, crumpled morphology with layered structure is visible (Figure 2B and Figure S3 (A)), originating from the thinness of MoS₂ atomic layers. In contrast, the exfoliated MoS₂ nanosheets are more flat and rigid (Figure S4 (A)). The thickness of representative $h-MoS_2$ and e-MoS₂ nanosheets are demonstrated to be 2~7 and 1-3 atomic layers, respectively; the interlayer spacing is ~0.65 nm (Figure 2C and Figure S4 (C)). The high-resolution TEM (HRTEM) image also reveals the well-defined hexagonal crystalline lattice of both the h-MoS₂ and e-MoS₂ nanosheets (Figure S3 (B) and Figure S4 (B)). X-ray photoelectron survey scan spectra (XPS) reveals the presence of Mo and S with atomic ratio of 1:2, and no any impurites in both h-MoS₂

and e-MoS₂ samples. (Figure S5 (A)). The high-resolution Mo 3d XPS spectra can be deconvoluted into two peaks of signals with binding energies of 228.4 and 231.5 eV for both samples, corresponding to the doublet $Mo3d_{5/2}$ and $Mo3d_{3/2}$, respectively (Figure S5 (B,C)). The high resolution S2p XPS spectra also shows two binding energies of 162.4 and 161.3 eV (h-MoS₂), corresponding to S2p_{1/2} and S2p_{3/2}, respectively (Figure S5 (D, E)). The crystalline structure of our MoS_2 nanosheets was further confirmed by the Raman spectra and X-ray diffraction (XRD) patterns. As it is shown in Figure S6, the Raman spectra have two peaks at ~381 and 406 cm^{-1} which are attributed to the A_{1g} and E_{2g} modes, respectively. The indexed XRD pattern is shown in Figure 2 (D), the peaks corresponding to the (100), (103), (110) planes are matching well the JCPDS card (37-1492). The weak (002) reflection indicates that the produced nanosheets are made of few layers assembled by van der Waals interactions (a spacing of 0.65 nm), in good agreement with the observation from TEM images.

The electrochemical performance of the as-prepared $h-MoS_2$ film based micro-supercapacitor was first studied by cyclic voltammetry (CV) at various scan rates over the voltage range of -0.5-0.5 V. As presented in **Figure 3**A and Figure S7 (see Supporting information), two hump peaks at

full papers



Figure 3. (A) CV curves of h-MoS₂ film based micro-supercapacitor at different scanning rates of 40, 100 and 200 mVs⁻¹. (B) Galvanostatic charge discharge (GAL) curves of h-MoS₂ micro-supercapacitor at different current densities from 0.22 to 0.45 A m⁻². (C) Galvanostatic cycling behavior at a current density of 0.22 A m⁻². (D) Cycling performance of h-MoS₂ micro-supercapacitor over 1000 cycles at a current density of 0.22 Am⁻², the inset of (D) is the cycle performance for the first 150 cycles.

~0.2 and -0.2 V are observed. With increasing the scan rate from 10 to 200 mV s⁻¹, both peaks become more dominant owing to the strengthened redox reaction ($MoS_2+K^++e^- = MoS-SK^+$), consistently with characteristics of MoS_2 electrodes in supercapcitors with aqueous electrolyte.^[25] This pseudo-capacitance behavior of h-MoS₂ microsupercapacitor is further confirmed by the nonlinear galvanostatic chargedischarge curves at different current densities as shown in Figure 3B. A high area capacitance of 8 mF cm⁻² is achieved at a scan rate of 10 mV s⁻¹, which is much higher than that reported for onion-like carbon- and graphene-based microsupercapatictors (0.5~3 mF cm⁻²).^[8,12]

The cycle performance of h-MoS₂ film based microsupercapacitor was further examined by charge-discharge testing at a current density of 0.22 Am⁻². Remarkably enough, only ~8% of capacitance fades after 1,000 cycles, representing excellent cycle stability. The small capacitance fading in the first 150 cycles perhaps originates from the "active" sites saturation on the surface of MoS₂ film during the charge-discharge process, and the later capacitance increase should be attributed to the "intercalation" induced capacitance behavior caused by the intercalation of foreign ions (Na⁺) into van der Waals gaps of multilayers of MoS₂. This phenomenon is similar to the one reported for most layered transition metal oxides (such as TiS₂, TiO₂ (B), MoO₃ etc) applied in energy storage.^[3,26-28]

In comparison, the liquid exfoliated MoS_2 nanosheets (e-MoS₂) prepared from bulk MoS_2 and commercially

available MoS₂ powders (size, $<2 \mu m$) were also employed for fabrication of MoS₂-based microsupercapacitors via the same procedures. (For details, see Supporting information). Apparently, the CV area of h-MoS₂ and e-MoS₂ is significantly higher than comparable bulk MoS₂ powder, indicating the higher capacitance for h-MoS₂ and e-MoS₂ based micro-supercapacitor (Figure 4A and Figure S7 and S8). This is further confirmed by their galvanostatic charge-discharge tests at a constant current density of 0.22 A m⁻². As presented in Figure 4B and Figure S6, the area capacitance of h-MoS₂ based micro-supercapacitor is 8 mF cm⁻², much higher than those of e-MoS₂ (~3.1 mF cm⁻²) and MoS₂ powder (1.8 mF cm⁻²). Correspondingly, the highest volumetric capacitance is 178 F cm⁻³ for h-MoS₂ based microsupercapacitor, which is comparable to those reported for the best micro-supercapacitors (~180 F cm⁻³ in organic electrolyte and ~160 F cm⁻³ in aqueous electrolyte).^[6] The superior electrochemical performance of h-MoS₂ film based microsupercapacitor should be attributed to the favorable structures of h-MoS₂ nanosheets for energy storage: (1) 2D graphene-like sheet structure, which maximizes the surface effects compared to bulk MoS₂ and restacked e-MoS₂ nanosheets, and provides a larger number of "active" sites for fast and reversible surface redox reactions; (2) rough morphology of MoS₂ layers, leading to a better pore structure for easy access of the ions from the aqueous electrolyte to the electrodes.



Figure 4. Comparison of the electrochemical performance of h-MoS₂ nanosheet, e-MoS₂ nanosheet and MoS₂ powder based micro-supercapacitors. (A) CV curves of h-MoS₂ nanosheet, e-MoS₂ nanosheet and MoS₂ powder based micro-supercapacitors at a scan rate of 100 mV s⁻¹. (B) Galvanostatic charge-discharge (GAL) of h-MoS₂ nanosheet, e-MoS₂ nanosheet and MoS₂ powder based micro-supercapacitors at the current density of 0.22 Am⁻². Comparisons of the (C) specific gravimetric capacitance and (D) volumetric capacitance of h-MoS₂ nanosheet, e-MoS₂ nanosheet and MoS₂ powder based micro-supercapacitors at various scan rates.

3. Conclusion

In summary, we have demonstrated a simple and scalable approach to fabricate film based micro-supercapacitors via spray painting of MoS_2 nanosheets and subsequent laser patterning. The optimized MoS_2 film based micro-supercapacitor exhibits high area (8 mF cm⁻²) and volumetric capacitances as well as excellent cycle performance, superior to EDLC-based micro-supercapacitors. Such efficient approach provides a good opportunity to develop various micro/nanosized energy storage devices to satisfy the requirements of portable, flexible, and transparent microelectronic devices.

4. Experimental Section

*Hydrothermal Method-Produced h-MoS*₂ *Nanosheets*: 10 mmol MoO₃, 25 mmol potassium thiocyanate and 64mg sodium dodecyl sulfate (SDS) were put into a Teflon-lined autoclave of 40 mL capacity, then, filled with 30 mL deionized water and sonicated for 30 min. The autoclave was sealed into a stainless steel tank and kept at 220 °C for 24 h, and then cooled to room temperature naturally. A dark powder was collected and washed with plenty of deionized water and surfactant. The remaining mixture is centrifuged and collected by vacuum filtration through a 0.45 μ m PVDF

filter membrane. The resulting solid is dried in vacuum oven at 90 $^{\circ}\text{C}$ overnight.

Liquid Exfoliation-Produced MoS_2 Nanosheets: Molybdenum disulfide powder was dissolved in DMF solvent and sonicated for 10 h (keep the water in sonicator cold). Centrifugation at 6000 rpm for 30 min was carried out and the supernatant was washed using ethanol and collected by pipette.

Characterization: Scanning electron microscopy (SEM) images were obtained on a high-resolution field-emission scanning electron microscope (FEI Quanta 400), the SEM samples were dissolved in ethanol and ultrasonicated for 30 min, following that dropped on a Si substrate for SEM testing. The TEM image and the selected area electron diffraction pattern were obtained on a JEOL 2100 Field Emission Gun TEM, the sample preparation process was similar to the one for SEM, but use the Cu grid with carbon supported instead of silicn substrate. XPS analyses were carried out on a PHI Quantera x-ray photoelectron spectrometer with a chamber pressure of 5×10^{-9} torr and an Al cathode as the X-ray source. The source power was set at 100 W, and pass energies of 140.00 eV for survey scans and 26.00 eV for high-resolution scans were used. Raman spectra were recorded with a Renishaw InVia Raman microscope using a 50x objective lens at room temperature, with a 514.5 nm laser beam and 1800 lines per mm grating. The Raman band of Si at 521 cm⁻¹ was used as a reference to calibrate the spectrometer. XRD data was collected on a Rigaku D/Max Ultima α Powder X-ray diffractometer, sample was put inside the zero background sample holder during the testing.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

This work was financially supported by the US Army Research Office through a MURI grant (W911NF-11-1-0362) on Novel Free-Standing 2D Crystalline Materials focusing on Atomic Layers of Nitrides, Oxides, and Sulfides.

- [1] J. H. Sung, S. J. Kim, S. H. Jeong, E. H. Kim, K. H. Lee, J. Power Sources 2006, 162, 1467–1470.
- [2] H. K. Kim, S. H. Choi, Y. S. Yoon, S. Y. Chang, Y. W. Ok, T. Y. Seong, *Thin Solid Films* **2005**, *475*, 54–57.
- [3] B. E. Conway, J. Electrochem. Soc. 1991, 138, 1539–1548.
- [4] G. A. Snook, P. Kao, A. S. Best, J. Power Sources 2011, 196, 1–12.
- [5] J. Wang, J. Polleux, J. Lim, B. Dunn, J. Phys. Chem. C 2007, 111, 14925–14931.
- [6] J. Chmiola, C. Largeot, P. L. Taberna, P. Simon, Y. Gogotsi, *Science* 2010, 328, 480–483.
- [7] D. Pech, M. Brunet, P. L. Taberna, P. Simon, N. Fabre, F. Mesnilgrente, V. Conedera, H. Durou, *J. Power Sources* 2010, 195, 1266–1269.
- [8] D. Pech, M. Brunet, H. Durou, P. H. Huang, V. Mochalin, Y. Gogotsi, P. L. Taberna, P. Simon, *Nat. Nanotechnol.* 2010, 5, 651–654.
- [9] V. L. Pushparaj, M. M. Shaijumon, A. Kumar, S. Murugesan, L. Ci, R. Vajtai, R. J. Linhardt, O. Nalamasu, P. M. Ajayan, *Proc. Natl. Acad. Sci. USA* **2007**, *104*, 13574–13577.
- [10] S. W. Lee, B. S. Kim, S. Chen, Y. Shao-Horn, P. T. Hammond, J. Am. Chem. Soc. 2009, 131, 671–679.
- [11] M. Kaempgen, C. K. Chan, J. Ma, Y. Cui, G. Gruner, *Nano Lett.* 2009, 9, 1872–1876.
- [12] W. Gao, N. Singh, L. Song, Z. Liu, A. L. M. Reddy, L. J. Ci, R. Vajtai, Q. Zhang, B. Q. Wei, P. M. Ajayan, *Nat. Nanotechnol.* **2011**, *6*, 496–500.
- [13] M. F. El-Kady, V. Strong, S. Dubin, R. B. Kaner, *Science* 2012, 335, 1326–1330.
- [14] Y. W. Zhu, S. Murali, W. W. Cai, J. W. Suk, J. R. Potts, R. S. Ruoff, *Adv. Mater.* 2010, 22, 3906–3924.
- [15] Y. W. Zhu, S. Murali, M. D. Stoller, K. J. Ganesh, W. W. Cai, P. J. Ferreira, A. Pirkle, R. M. Wallace, K. A. Cychosez, M. Thommes, D. Su, E. A. Stach, R. S. Ruoff, *Science* **2011**, *332*, 1537–1541.

- [16] B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti, A. Kis, Nat. Nanotechnol. 2011, 6, 147–150.
- [17] B. Radisavljevic, M. B. Whitwick, A. Kis, ACS Nano 2011, 5, 9934–9938.
- [18] S. Bertolazzi, J. Brivio, A. Kis, ACS Nano 2011, 5, 9703-9 709.
- [19] D. J. Late, B. Liu, H. S. S. R. Matte, V. P. Dravid, C. N. R. Rao, ACS Nano 2012, 6, 5635–5641.
- [20] Y. Y. Peng, Z. Y. Meng, C. Zhong, J. Lu, W. C. Yu, Y. B. Jia, Y. T. Qian, *Chem. Lett.* **2001**, *8*, 772–773.
- [21] C. N. R. Rao, A. Nag, Eur. J. Inorg. Chem. 2010, 27, 4244– 4250.
- [22] H. S. S. R. Matte, A. Gomathi, A. K. Manna, R. Datta, S. K. Pati, C. N. R. Rao, Angew. Chem., Int. Ed. 2010, 49, 4059–4062.
- [23] J. N. Coleman, M. Lotya, A. O'Neill, S. D. Bergin, P. J. King, U. Khan, K. Young, A. Gaucher, S. De, R. J. Smith, I. V. Shvets, S. K. Arora, G. Stanton, H. Y. Kim, K. Lee, G. T. Kim, G. S. Duesberg, T. Hallam, J. J. Boland, J. J. Wang, J. F. Donegan, J. C. Grunlan, G. Moriarty, A. Shmeliov, R. J. Nicholls, J. M. Perkins, E. M. Grieveson, K. Theuwissen, D. W. McComb, P. D. Nellist, V. Nicolosi, *Science* 2011, *331*, 568–571.
- [24] R. J. Smith, P. J. King, M. Lotya, C. Wirtz, U. Khan, S. De, A. O'Neill, G. S. Duesberg, J. C. Grunlan, G. Moriarty, J. Chen, J. Z. Wang, A. I. Minett, V. Nicolosi, J. N. Coleman, *Adv. Mater.* **2011**, *23*, 3944.
- [25] J. M. Soon, K. P. Loh, Electrochem. Solid-State Lett. 2007, 10, A250–A254.
- [26] B. E. Conway, *Electrochim. Acta* **1993**, *38*, 1249–1258.
- [27] T. Brezesinski, J. Wang, S. H. Tolbert, B. Dunn, Nat. Mater. 2010, 9, 146–151.
- [28] M. Zukalova, M. Kalbac, L. Kavan, I. Exnar, M. Graetzel, *Chem. Mater.* 2005, 17, 1248–1255.
- [29] Z. Y. Yin, H. Li, H. Li, L. Jiang, Y. M. Shi, Y. H. Sun, G. Lu, Q. Zhang, X. D. Chen, H. Zhang, ACS Nano 2012, 6, 74–80.
- [30] Z. Y. Zeng, Z. Y. Yin, X. Huang, H. Li, Q. Y. He, G. Lu, F. Boey, H. Zhang, *Angew. Chem.*, *Int. Ed.* **2011**, *50*, 11093– 11097.
- [31] K. K. Liu, W. J. Zhang, Y. H. Lee, Y. C. Lin, M. T. Chang, C. Su, C. S. Chang, H. Li, Y. M. Shi, H. Zhang, C. S. Lai, L. J. Li, *Nano Lett.* 2012, *12*, 1538–1544.
- [32] H. Li, Z. Y. Yin, Q. Y. He, H. Li, X. Huang, G. Lu, D. W. H. Fam, A. I. Y. Tok, Q. Zhang, H. Zhang, *Small* **2012**, *8*, 63–67.
- [33] Q. Y. He, Z. Y. Zeng, Z. Y. Yin, H. Li, S. X. Wu, X. Huang, H. Zhang, Small 2012, 8, 2994–2999.
- [34] J. Q. Liu, Z. Y. Zeng, X. H. Cao, G. Lu, L. H. Wang, Q. L. Fan, W. Huang, H. Zhang, *Small* **2012**, *8*, 3517–3522.
- [35] W. J. Zhou, Z. Y. Yin, Y. P. Du, X. Huang, Z. Y. Zeng, Z. X. Fan, H. Liu, J. Y. Wang, H. Zhang, *Small* **2013**, *9*, 140–147.

Received: December 17, 2012 Revised: January 21, 2013 Published online: April 16, 2013